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# 单晶体 α-NaYF<sub>4</sub>: Dy<sup>3+</sup>的制备及光谱特性

姜永章1,夏海平1,张加忠1,杨硕1,江东升1,王成1,冯治刚1,张健1,

谷雪梅1,章践立1,江浩川2,陈宝玖3

(1 宁波大学 光电子功能材料重点实验室, 浙江 宁波 315211)

(2 中国科学院宁波材料技术与工程研究所,浙江 宁波 315211)

(3 大连海事大学 物理系, 辽宁 大连 116026)

摘 要:以氯化钾(KF)作为助熔剂,选用 70~90℃/cm 的固液界面温度梯度,通过改进的助熔剂-布里 奇曼方法生长出了 Dy<sup>3+</sup> 离子掺杂的  $\alpha$ -NaYF4 单晶体.测试了样品的 X 射线粉末衍射,吸收光谱,激发光 谱,以及在不同紫外光激发下的发射光谱,分析了样品的的物相和光谱特性.应用色度学理论计算了 Dy<sup>3+</sup> 离子掺杂  $\alpha$ -NaYF4 单晶的色度坐标和色温.结果表明:当原料组份配比为 30NaF-18KF-52YF3 时,有利于大尺寸晶体生长.原料中 KF 的加入起到助熔剂作用,它改变了 NaF-YF3 二元系相图并且降低 了  $\alpha$ -NaYF4 的熔点,使得  $\alpha$ -NaYF4 能从熔融的 NaF-KF-YF3 混合物中结晶出来,最后生长出透明的  $\alpha$ -NaYF4 单晶体.实验发现大多 KF 被排析在生长后期的晶体顶端,KF 的加入没有对  $\alpha$ -NaYF4 单晶体的 结构产生影响.在 348 nm 紫外光激发下,单晶体可发射出蓝光约 479 nm,黄光约 571 nm 以及弱的红光 约 659 nm.其中 1.299 mol% Dy<sup>3+</sup>掺杂浓度的  $\alpha$ -NaYF4 单晶体发射的蓝、黄与红光可耦合出色坐标 x = 0.285, y = 0.338 以及色温为 8 065 K 的白光.这些新型材料在紫外激发的白光二极管领域具有潜在的应用.

**关键词**:光学材料;光谱;布里奇曼方法;α-NaYF<sub>4</sub>单晶体;白光发射;紫外光 **中图分类号**:O782;O734 **文献标识码**:A **文章编号**:1004-4213(2015)08-0816001-6

# Growth and Optical Spectra of Dy<sup>3+</sup> Doped α-NaYF<sub>4</sub> Single Crystal

JIANG Yong-zhang<sup>1</sup>, XIA Hai-ping<sup>1</sup>, ZHANG Jia-zhong<sup>1</sup>, YANG Shuo<sup>1</sup>, JIANG Dong-sheng<sup>1</sup>, WANG Cheng<sup>1</sup>, FENG Zhi-gang<sup>1</sup>, ZHANG Jian<sup>1</sup>, GU Xue-mei<sup>1</sup>, ZHANG Jian-li<sup>1</sup>, JIANG Hao-chuan<sup>2</sup>, CHEN Bao-jiu<sup>3</sup>

(1 Ningbo University, Key laboratory of Photo-electronic Materials, Ningbo, Zhejiang 315211, China)

(2 The Chinese Academy of Sciences, Ningbo Institute of Materials Technology and Engineering,

Ningbo, Zhejiang 315211, China)

(3 Dalian Maritime University, Department of Physics, Dalian, Liaoning 116026, China)

Abstract: The  $Dy^{3+}$  doped  $\alpha$ -NaYF<sub>4</sub> single crystals were synthesized by a modified Bridgman method under the conditions of taking KF as an assistant flux and a temperature gradient (70~90 °C/cm) at the solid-liquid interface. X-ray diffraction, absorption spectra, excitation spectra, and emission spectra were measured to investigate the phase and spectrum properties of the crystals. The chromaticity coordinates and color temperature of crystals were calculated based on Colorimetry theories. Many growing experiments demonstrate that the chemical compositions of  $30NaF-18KF-52YF_3$  (molar ratio) are

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First author: JIANG Yong-zhang (1991-), male, M. S. degree candidate, mainly focuses on LED material technology research, and function single crystal growth. Email:1373772919@qq. com

Supervisor(Contact author): XIA Hai-ping (1967-), male, professor, Ph. D. degree, mainly focuses on optical material synthesis, functional crystal growth research, laser optics, etc. Email:hpxcm@nbu.edu.cn

beneficial to obtain big  $\alpha$ -NaYF<sub>4</sub> single crystals. The introduction of KF changed the phase equilibrium between NaF and YF<sub>3</sub> completely and lowered the melting point of  $\alpha$ -NaYF<sub>4</sub>, resulting into crystallization of  $\alpha$ -NaYF<sub>4</sub> just from the mixture when it is cooled. Most of KF was excluded on the top of crystals which grew at the final stage of crystal growth. The introduction of flux KF did not change the structure of  $\alpha$ -NaYF<sub>4</sub> single crystals. The emission blue 479 nm, yellow 571 nm and weak red 659 nm could be obtained simultaneously under the excitation of ultraviolet lights. A white light emission from the coupling of the above three lights with chromaticity coordinates of x = 0.285, y = 0.338, color temperature  $T_c = 8~065$  K could be obtained from 1.299 mole% Dy<sup>3+</sup> doped  $\alpha$ -NaYF<sub>4</sub> single crystal when excited by a 348 nm light. This indicates that Dy<sup>3+</sup> doped  $\alpha$ -NaYF<sub>4</sub> single crystals can be a potential candidate for white light emitting diode excited by ultraviolet light.

Key words: Optical material; Optical spectrum; Bridgman method;  $\alpha$ -NaYF<sub>4</sub> single crystal; White light emission; Ultraviolet light

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## 0 Introduction

Recently, the rare-earth ions doped single-host full-color emitting solid state materials for white Light Emitting Diode (w-LED) application have attracted a significantly increasing attention because they exhibit favorable characteristics such as high brightness, high luminous efficiency, low light scattering and low power consumption. These advantages could overcome the drawbacks of using the common two methods to achieve a white light emission<sup>[1-3]</sup>. The w-LED now is an important class of lighting devices. It exhibits a high potential for replacement of conventional lighting sources such as incandescent and fluorescent lamps, and will be applied widely in the future<sup>[1]</sup></sup>. So far, a lot of works have been carried out on the possible application of rare earth ions doped single-host fullcolor emitting materials like glasses, glass ceramics for w-LED<sup>[4-6]</sup>.

As is well known, the host materials have also been playing a significant role in lights and w-LED devices. Previous work mainly focused on the glasses and glass ceramics. There are scarce reports on rare earth ions doped single crystals for w-LED due to the difficulty in crystal growth. In fact, many single crystals show a good thermal, mechanical and chemical stability, and a high luminous efficiency of rare earth ions due to their rigid cyclic symmetric structure. The excellent chemical-physical and optical properties of single crystals are favorable for them to be applied in LED device with high stability, long lifetime and high luminous efficiency. In recent years, fluoride single crystals have become the most investigated efficient luminescent material for their low phonon energy. More recently, white light emission from  $Tb^{3+}/Sm^{3+}$ co-doped LiYF4 single crystals was realized in our laboratory [7]. Indeed, some other fluoride single crystals such as  $NaYF_4$  existed two structures<sup>[8-9]</sup>, one being cubic  $\alpha$ -NaYF<sub>4</sub> and the other hexagonal  $\beta$ -NaYF<sub>4</sub>

were used as optical materials<sup>[10-12]</sup>. Compared with other single crystals such as LiYF<sub>4</sub> crystals  $(\sim 650 \text{ cm}^{-1})^{[13]}$ ,  $\alpha$ -NaYF<sub>4</sub> single crystals have a lower phonon energy ( $\sim 360 \text{ cm}^{-1}$ ) and a higher transimitance in UV to visible light region<sup>[10]</sup>, which are favorable for their appplication in w-LED.

Rare earth ions generally possess of plenty of wellshield 4f states that could emit emission covering different wavelength from ultraviolet to infrared. As a lanthanide, Dy<sup>3+</sup> ions have transitions in blue region near 479 nm ( ${}^{4}F_{9/2} \rightarrow {}^{6}H_{15/2}$ ), yellow light region near 571 nm ( ${}^{4}F_{9/2} \rightarrow {}^{6}H_{13/2}$ ) and red light region near 659 nm ( ${}^{4}F_{9/2} \rightarrow {}^{6}H_{11/2}$ ) when excited by a suitable ultraviolet light<sup>[6,14-20]</sup>. These emissions including blue, yellow, and red light emitting simultaneously could potentially enable a good white light emission. It has been reported that Dy3+ doped solid state materials such as  $phosphors^{[14-18]}$ ,  $glasses^{[19]}$  and single crystals<sup>[20]</sup> for white LED application. Compared with doubly and triply doped materials, singly doped one could reduce unnecessary energy consumption in the process of energy transfer among two or three rare earths in host materials.

In this work,  $Dy^{3+}$  doped  $\alpha$ -NaYF<sub>4</sub> single crystals were obtained by an improved Bridgman method. The capability of generating white light by simultaneous emissions of red, yellow and blue emitting fluorescent centers under the excitation of ultraviolet light has been demonstrated.

## **1** Experiment and measurement

The Dy<sup>3+</sup> doped  $\alpha$ -NaYF<sub>4</sub> single crystals were successfully grown by a vertical Bridgman method. The raw materials (NaF, KF, YF<sub>3</sub>, DyF<sub>3</sub> powders) with a 99.99% purity were weighed according to the molar ratio of 30NaF-18KF-51. 2YF<sub>3</sub>-0. 8DyF<sub>3</sub> respectively. An excessive KF as a flux was critical for obtaining transparent Dy<sup>3+</sup> doped  $\alpha$ -NaYF<sub>4</sub> single crystals in process of crystal growth. Our experiments demonstrated that the introduction of the fluoride salt KF changed the phase equilibrium between NaF and YF<sub>3</sub> completely and lowered the melting point of  $\alpha$ -NaYF<sub>4</sub>, resulting into crystallization of  $\alpha$ -NaYF<sub>4</sub> just from the mixture when it is cooled. And then the mixture of those materials was ground about 1 h in a mortar, the moisture and some oxygen impurities in the fluoride powders could be removed completely using the high temperature hydrofluorination method by which the powders were sintered with anhydrous HF at 750 °C for 8 h. The seed from previous work was obtained by spontaneous nucleation from the seed wells and oriented along a-axis. The seed was put in the bottom of seed well and then the polycrystalline powders were filled in Pt crucibles with a size of  $\Phi$ 10 mm $\times$ 150 mm. The temperature gradient at solid-liquid interface was  $70 \sim 90$  °C/cm, The growth process was carried on by lowering the crucible at a rate of  $\sim$ 0.6 mm/h. The detailed process is similar to the report of growth of  $LiYF_4$  single crystals<sup>[21]</sup>. The initial and middle parts of boule are transparent, while the final part of the boule is white opaque which is KF and some deviating components excluded from melt. The as-grown crystal was cut into pieces and well polished to  $\sim 2.0$  mm thickness. One is displayed in the inset of Fig. 1.



Fig. 1 The XRD pattern of the grown single crystal and standard pattern of the  $\alpha$ -NaYF<sub>4</sub> (JCPD 77-2042). The insert is the photo of polished crystal piece D<sub>0</sub>

The Dy<sup>3+</sup> doping concentrations in  $\alpha$ -NaYF<sub>4</sub> single crystals were measured by the Inductively Coupled atomic Plasma (ICP) emission spectrometry (PerkinElmer Inc., Optima 3000). The 1.292 mole% and 1.299 mole% of  $Dy^{3+}$  ions were determined for the lower part of  $\alpha$ -NaYF<sub>4</sub> single crystals (designated D0), which was corresponding to the initial stage of growing process, and the upper part (designated  $D_1$ ) to the of А final stage growing process. XD-98X diffractometer (XD-3, Beijing) was used to identify the phase composition of the crystal. The Cary 5000 UV/ VIS/NIR spectrophotometer (Agilent Co., America) recorded the absorption spectra ranging from 240 nm to 600 nm. The optical properties of the  $\alpha$ -NaYF<sub>4</sub> single

crystal doped with  $Dy^{3+}$  ions were characterized by excitation spectra and photoluminescence spectra measured on a F-4500 spectrophotometer (Hitachi high-technologies Co., Tokyo Japan). All the measurements were carried out at room temperature, and kept the same conditions for the samples in order to get comparable results. The chromaticity coordinates and color temperature of the samples were calculated from the emission spectra.

# 2 **Results and discussion**

#### 2.1 Phase composition analysis

Figure 1 shows the XRD patterns of the lower and upper parts of obtained single crystals which are corresponding to the initial and final stages of crystal growth respectively, as well as standard card of  $\alpha$ -NaYF<sub>4</sub>. The visible peaks and its corresponding lattice plane indexes were also marked in Fig. 1. Comparing with reference card JCPD  $\ddagger$  77-2042<sup>[9,12]</sup>, the diffraction peaks of both lower and upper parts of  $\alpha$ -NaYF4 single crystals doped with Dy3+ ions are well matched with those of standard  $\alpha$ -NaYF<sub>4</sub>, indicating that this transparent crystal has pure cubic phase. It confirms that the current Dy3+ ions doping level and introducing of K<sup>+</sup> ions do not bring any meaningful change for single crystals structure composition. As is well known,  $\alpha$ -NaYF<sub>4</sub> single crystals have fluorite structure ( $CaF_2$ ), in which  $Na^+$  and  $Y^{3+}$  ions are randomly distributed in the cation sites where are situated in cubic center, and its lattice constants could be calculated from the XRD pattern data as following lattice parameters:  $a_{\rm L} = b_{\rm L} = c_{\rm L} = 0.5510$  nm,  $a_{\rm U} = b_{\rm U} =$  $c_{\rm U} = 0.556$  2 nm for the lower and upper part crystal respectively, which are close to the standard parameters as shown in Fig. 2.





#### 2.2 Phase composition analysis

Fig. 2 displays the absorption spectra of the lower part, upper part and undoped  $\alpha$ -NaYF<sub>4</sub> single crystals which are labeled curves a, b and c respectively, and cover from 240 nm to 600 nm. It is obvious that undoped  $\alpha$ -NaYF<sub>4</sub> single crystals have no visible absorption peaks referring to curve *c*. Moreover, curve *a* and *b* have the similar absorption peaks while the absorption intensity is different from each other. Five absorption peaks located at ~325 nm, ~348 nm, ~364 nm, ~387 nm and ~452 nm are labeled in Fig. 2, which represent the absorptive transitions of Dy<sup>3+</sup> ions in  $\alpha$ -NaYF<sub>4</sub> single crystal from stable state (<sup>6</sup> H<sub>15/2</sub>) to excitation states (<sup>6</sup> P<sub>3/2</sub>, <sup>4</sup> M<sub>17/2</sub>), (<sup>6</sup> P<sub>7/2</sub>, <sup>4</sup> I<sub>11/2</sub>), (<sup>6</sup> P<sub>5/2</sub>, <sup>4</sup> P<sub>3/2</sub>), (<sup>4</sup> I<sub>13/2</sub>, <sup>4</sup> F<sub>7/2</sub>) and <sup>4</sup> I<sub>15/2</sub>. It is proved that Dy<sup>3+</sup> doped  $\alpha$ -NaYF<sub>4</sub> single crystals could be excited efficiently by ultraviolet light or blue light.

#### 2.3 Excitation spectra analysis

The energy level transition rules could be concluded from Ref. 13 to Ref. 19 about  $Dy^{3+}$  ions and are shown in Fig. 3. It is implied that  $Dy^{3+}$  ions in the suitable host material could emit the blue, yellow and red light simultaneously under the ultraviolet excitation and that could enable a white light emission for LED application.





To investigate the effect of excitation wavelengths on the emission spectra, we analyze the excitation spectra of the  $Dy^{3+}$  doped  $\alpha$ -NaYF<sub>4</sub> single crystals. Fig. 4 exhibits the excitation spectra of the lower part crystal monitored at 479 nm, 571 nm and 659 nm which are labeled *d*, *e* and *f* respectively. There are four excitation peaks at 324 nm ( ${}^{6}H_{15/2} \rightarrow {}^{4}M_{17/2}$ ,  ${}^{6}P_{3/2}$ ), 348 nm ( ${}^{6}H_{15/2} \rightarrow {}^{6}P_{7/2}$ ,  ${}^{4}I_{11/2}$ ), 364 nm ( ${}^{6}H_{15/2}$  $\rightarrow {}^{6}P_{5/2}$ ,  ${}^{4}P_{3/2}$ ) and 387 nm ( ${}^{6}H_{15/2} \rightarrow {}^{4}I_{13/2}$ ,  ${}^{4}F_{7/2}$ ) for the spectrum *d* and *e*. Thus, the excitation peaks at 324 nm, 348 nm, 364 nm and 387 nm are proper to be chosen as excitation lights.





Fig. 5 and Fig. 6 display the emission spectra of the 1. 292 mole $\frac{1}{2}$  /1. 299mole $\frac{1}{2}$  Dy<sup>3+</sup> doped NaYF<sub>4</sub> crystal 348, 364, 387 under 324, nm wavelengths, respectively. When excited by these excitation wavelengths, the emission spectra of  $\alpha$ -NaYF<sub>4</sub> single crystals doped with Dy3+ ions could emit a stronger blue light at  $\sim 479$  nm, corresponding to the magnetic dipole transition (  $^4\,F_{_{9/2}} \rightarrow ^6 H_{_{15/2}}$  ), a yellow light at  $\sim$ 571 nm, related to the electric dipole transition (  $^4F_{_{9/2}}{\twoheadrightarrow}^6H_{_{13/2}})$  , and a weaker red light at  $\sim659$  nm about the  ${}^{4}F_{9/2} \rightarrow {}^{6}H_{11/2}$  transition respectively. As is well-known, the magnetic dipole transition is insensitive to the crystal field surrounding  $\mathrm{Dy}^{3+}$  ions while the electric dipole transition is strongly influenced by the crystal environment of  $Dy^{3+}$  ions. When  $Dy^{3+}$ ions are situated in a high symmetry site, the blue light emission would be stronger than the yellow one. In reverse, the yellow light emission is dominant in the emission<sup>[6,13,15]</sup>. Thus, the stronger blue light emission of  $Dy^{3+}$  ions doped  $\alpha$ -NaYF<sub>4</sub> single crystals from those emission spectra indicates that Dy<sup>3+</sup> ions occupied a high symmetry site in  $\alpha$ -NaYF<sub>4</sub> host. The emission spectra also implied that the possibility of a white light



Fig. 5 The emission spectra and luminous parameters of \$1.292 mole%\$ Dy3+ doped  $NaYF_4$  crystal under various excitation wavelengths



Fig. 6 The emission spectra and luminous parameters of  $1.\,299\,mole\,\%~Dy^{3+}$  doped  $NaYF_4$  crystal under various excitation wavelengths

emission from the combination of above blue, yellow and red lights could be come true in  $Dy^{3+}$  ions doped  $\alpha$ -NaYF<sub>4</sub> single crystals.

#### 2.5 CIE chromaticity coordinates

It is necessary to mark chromaticity coordinates of the samples on standard chromaticity diagram for reflecting their luminescence color, the chromaticity coordinates for the emissions of the  $Dy^{3+}$  doped  $NaYF_4$ crystal under various excitation wavelengths can be calculated using the following formula<sup>[22]</sup>

$$x = \frac{X}{X+Y+Z}, y = \frac{Y}{X+Y+Z}, z = \frac{Z}{X+Y+Z}$$
(1)

Where X, Y and Z are three tristimulus values. Those three values are given by the following formula<sup>[22]</sup>

$$X = \int_{\lambda} P(\lambda) \bar{x}(\lambda) d\lambda$$
  

$$Y = \int_{\lambda} P(\lambda) \bar{y}(\lambda) d\lambda$$
 (2)  

$$Z = \int_{\lambda} P(\lambda) \bar{z}(\lambda) d\lambda$$

Where  $\lambda$  is the wavelength of the equivalent monochromatic light.  $P(\lambda)$  is spectral power distribution.  $\bar{x}(\lambda)$ ,  $\bar{y}(\lambda)$  and  $\bar{z}(\lambda)$  are the three color matching functions.

Color temperature ( $T_c$ ) is also a key technical factor for evaluating the applicability of a luminescence material. Several methods for calculating  $T_c$  have been published<sup>[23-25]</sup>. The McCamy<sup>[23]</sup> method only provides a maximum absolute error of less than 2 degrees Kelvin for color temperatures below 5 000 K. The method of Qiu Xing-zhong<sup>[24]</sup> which involves different equations in two ranges of an angle in u, v space, is best, but it is more complicated than former formula. In this paper, the  $T_c$  can be calculated by software (Chromaticity Coordinates To Color Temperature Conversion).

The chromaticity coordinates of the lower part, upper part crystals under various excitation wavelengths were calculated using Eq.  $(1) \sim (2)$  and they are listed in Fig. 5, Fig. 6 respectively. These

chromaticity coordinates were marked in CIE chromaticity coordinate diagrams exhibited in Fig. 7 and Fig. 8. It could be found that the emission light under a 348 nm excitation wavelength falls into the white light region. Comparing Fig. 7 with Fig. 8, we could find that the emission of the upper part crystal under a 348 nm excitation wavelength is much closer to the ideal coordinate. Although the white emission could be achieved, more effort should be paid to decrease the color temperature of  $\alpha$ -NaYF<sub>4</sub> single crystals doped with Dy<sup>3+</sup> ions, which is relegated to further work.









# **3** Conclusions

As a whole,  $Dy^{3+}$  doped  $\alpha$ -NaYF<sub>4</sub> single crystals could be synthesized by a modified Bridgman method. The KF was found to be a very favorable assistant flux to grow  $\alpha$ -NaYF<sub>4</sub> single crystal in NaY-YF<sub>3</sub> system and the Dy<sup>3+</sup> ion was easy to institute for Y<sup>3+</sup> ion due to the same value state and similar size of radii. Strong emissions at blue ~ 479 nm, yellow ~ 571 nm and weak emission at red ~ 659 nm could be obtained simultaneously under excitation of ultraviolet lights. A white light emission from the coupling of the above three lights with chromaticity coordinates of x=0.285, y=0.338, color temperature  $T_c = 8~065$  K could be obtained from 1. 299 mole% Dy<sup>3+</sup> doped  $\alpha$ -NaYF<sub>4</sub> single crystal when excited by a 348 nm light. And the chromaticity coordinate and color temperature of the crystal was varied slightly with the excitation wavelength or adjusting the concentration of  $Dy^{3+}$  ion in  $\alpha$ -NaYF<sub>4</sub> crystals. Due to the chemical stability, high luminous efficiency and unique white light-emitting of  $Dy^{3+}$  doped  $\alpha$ -NaYF<sub>4</sub> single crystals, it can be a potential candidate for white LED application. **References** 

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